

# Room-temperature superconductivity in a carbonaceous sulfur hydride

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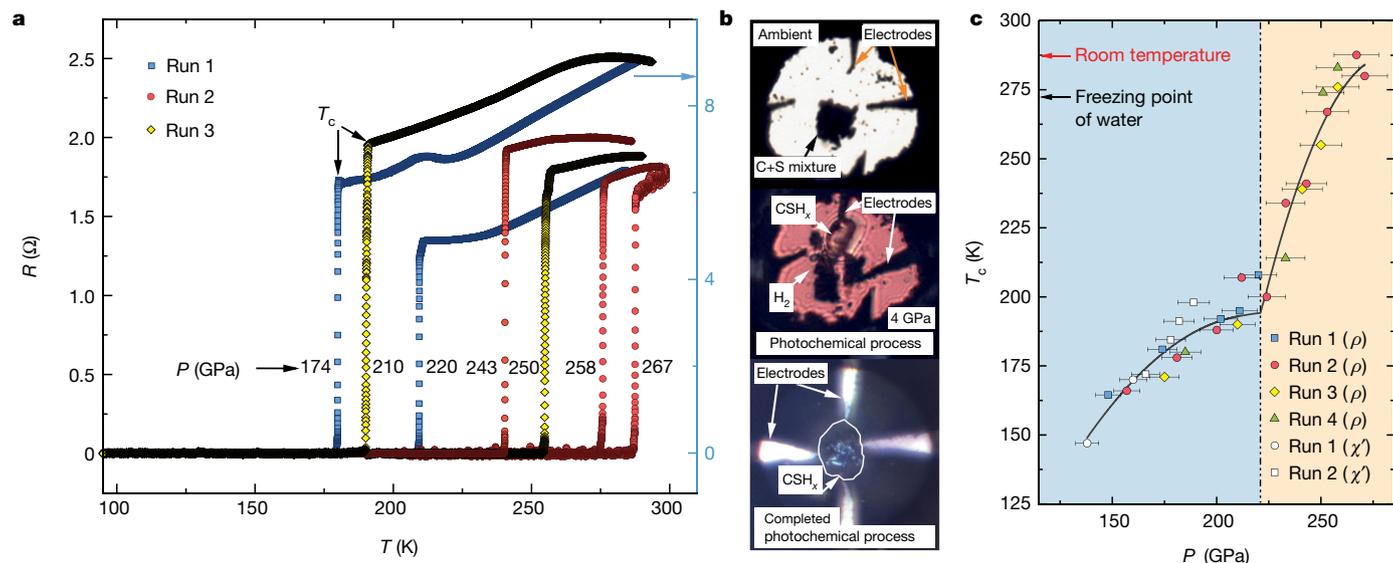
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One of the long-standing challenges in experimental physics is the observation of room-temperature superconductivity<sup>1,2</sup>. Recently, high-temperature conventional superconductivity in hydrogen-rich materials has been reported in several systems under high pressure<sup>3–5</sup>. An important discovery leading to room-temperature superconductivity is the pressure-driven disproportionation of hydrogen sulfide (H<sub>2</sub>S) to H<sub>3</sub>S, with a confirmed transition temperature of 203 kelvin at 155 gigapascals<sup>3,6</sup>. Both H<sub>2</sub>S and CH<sub>4</sub> readily mix with hydrogen to form guest–host structures at lower pressures<sup>7</sup>, and are of comparable size at 4 gigapascals. By introducing methane at low pressures into the H<sub>2</sub>S + H<sub>2</sub> precursor mixture for H<sub>3</sub>S, molecular exchange is allowed within a large assemblage of van der Waals solids that are hydrogen-rich with H<sub>2</sub> inclusions; these guest–host structures become the building blocks of superconducting compounds at extreme conditions. Here we report superconductivity in a photochemically transformed carbonaceous sulfur hydride system, starting from elemental precursors, with a maximum superconducting transition temperature of 287.7 ± 1.2 kelvin (about 15 degrees Celsius) achieved at 267 ± 10 gigapascals. The superconducting state is observed over a broad pressure range in the diamond anvil cell, from 140 to 275 gigapascals, with a sharp upturn in transition temperature above 220 gigapascals. Superconductivity is established by the observation of zero resistance, a magnetic susceptibility of up to 190 gigapascals, and reduction of the transition temperature under an external magnetic field of up to 9 tesla, with an upper critical magnetic field of about 62 tesla according to the Ginzburg–Landau model at zero temperature. The light, quantum nature of hydrogen limits the structural and stoichiometric determination of the system by X-ray scattering techniques, but Raman spectroscopy is used to probe the chemical and structural transformations before metallization. The introduction of chemical tuning within our ternary system could enable the preservation of the properties of room-temperature superconductivity at lower pressures.

In the past decade there has been an emergence of interest in the discovery of materials relevant to room-temperature superconductivity. Extreme pressure has already been proven to be the most versatile order parameter because it facilitates the production of new quantum materials with unique stoichiometries and a mechanism for pressure-induced metallization<sup>8–10</sup>. This has been most essential for non-metallic starting materials<sup>11–13</sup>. All systems with high superconducting critical temperature ( $T_c > 200$  K) that have been accessed under pressure so far are hydrogen-rich materials, in which the superconductivity is driven by strong electron–phonon coupling to high-frequency hydrogen phonon modes<sup>14,15</sup>. However, the specific stoichiometry (that is, XH<sub>n</sub>) does not seem to be as critical as having a hydrogen-rich chemical environment that mimics the properties (electron density near/at the

Fermi surface and high-frequency phonon modes) of idealized pure metallic hydrogen<sup>16–19</sup>. This is highlighted by the difference between purely covalent systems such as H<sub>2</sub>S compared to metal hydride systems. The most recent example of a metal hydride is lanthanum hydride (LaH<sub>10±x</sub>), which has  $T_c = 250–260$  K at 180–200 GPa (refs. 4,5). A lanthanum ‘superhydride’ has been experimentally realized, although a precise determination of its stoichiometry is lacking, as are the tools to determine such parameters. Investigation of the predicted band structures of rare-earth (La and Y) superhydrides implies an ionic heavy atom that donates its valence electrons to the hydrogen network, stabilizing a clathrate-like hydrogen cage structure<sup>20–22</sup>. Despite the large number of theoretical predictions for possible hydrogen-rich materials at high pressure, very few demonstrate superconducting behaviour at

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**Fig. 1 | Superconductivity in C–S–H at high pressures.** **a**, Temperature-dependent electrical resistance of the C–S–H system at high pressures ( $P$ ), showing superconducting transitions at temperatures as high as  $287.7 \pm 1.2$  K at  $267 \pm 10$  GPa. The data were obtained during the warming cycle to minimize the electronic and cooling noise. We note that the left and right vertical axes represent results from two different experimental runs. **b**, microphotographs showing the photochemical process of superconducting C–S–H sample with electrical leads in a four-probe configuration for resistance measurements. **c**, Pressure dependence of  $T_c$ , as determined by the sharp drop in the electrical

resistance ( $\rho$ ) and a.c. susceptibility ( $\chi'$ ) measurements shown in Figs. 1a, 2a.  $T_c$  increases with pressure from  $\sim 140$  GPa, then gradually levels off to  $\sim 194$  K around 220 GPa, and then sharply increases afterwards, showing a discontinuity around 225 GPa. The highest  $T_c$  observed was  $287.7$  K at 267 GPa. The low-temperature quasi-four-point resistance measurement at 271 GPa (the highest pressure measured) shows a superconducting transition at  $\sim 280$  K. The solid lines are to guide the eye and different colours represent different experiments. The red and black arrows represent room temperature ( $15^\circ\text{C}$ ) and the freezing point of water, respectively. Error bars reflect uncertainty in the measured value.

high transition temperatures. This comes from either the predicted material not having the ideal chemical environment for H, or from the limitations of standard density functional theory tools to account for anharmonicity and for the quantum nature of H (ref. <sup>23</sup>).

Covalent metals present an alternative path to realizing room-temperature superconductivity, with the superconductivity of the exemplary system of  $\text{MgB}_2$  being driven by strongly covalent-bonding/antibonding states crossing the Fermi energy<sup>24</sup>. Covalent hydrogen-rich organic-derived materials are another class of high- $T_c$  materials that combine the advantages of covalent metals and metal superhydrides<sup>25,26</sup>; an example is  $\text{H}_3\text{S}$  (refs. <sup>3,27</sup>). Interest in these materials has been long-standing since Little's proposal of superconductivity at room temperatures in one-dimensional organic polymers with highly polarizable side chains<sup>28</sup> and Ginzburg's model of two-dimensional alternating conducting/dielectric 'sandwich' layers<sup>2,29</sup>. The removal of the heavy metal from superstoichiometric hydrides in covalent hydrogen-rich systems offers a promise for 'greener' future materials synthesized using low-cost, earth-abundant organic reactants. Here, we report superconductivity in a simple organic-derived C–S–H system with a highest  $T_c$  of about 288 K over a large pressure range between  $\sim 140$  GPa and  $\sim 275$  GPa, characterized by electrical resistance, magnetic susceptibility and field-dependence electrical transport measurements, as well as Raman spectroscopy. A series of structural and electronic phase transitions from molecular to metallic and superconducting are confirmed.

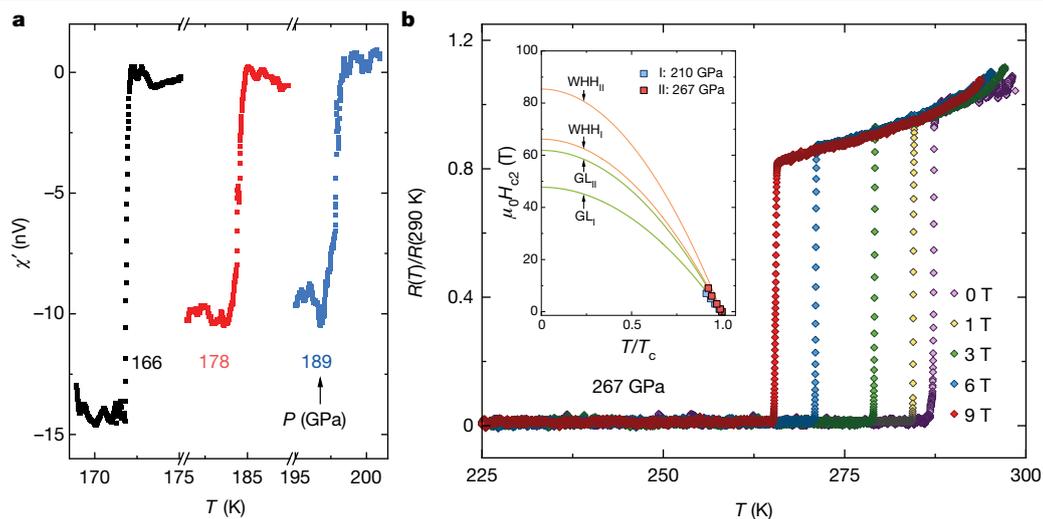
## Superconductivity in carbonaceous sulfur hydride

The photochemically synthesized C–S–H system becomes superconducting with its highest critical temperature being  $T_c = 287.7 \pm 1.2$  K at  $267 \pm 10$  GPa. The temperature probe's accuracy is  $\pm 0.1$  K. The superconducting transition was evidenced by a sharp drop in resistance towards zero for a temperature change of less than 1 K (Fig. 1a), which was measured

during the natural warming cycle ( $\sim 0.25$  K  $\text{min}^{-1}$ ) from low temperature with a current of  $10 \mu\text{A}$ – $1$  mA. The transition temperature determined from the onset of superconductivity appears to be approaching a dome shape as a function of pressure (Fig. 1b). It increases from 147 K at  $138 \pm 7$  GPa until it levels off to  $\sim 194$  K at about 220 GPa, with the pressures measured from the diamond edge using the Akahama 2006 scale<sup>30</sup> and calibrated  $\text{H}_2$  vibron frequency (see Methods). Remarkably, a sharp increase in  $T_c$  is observed above 220 GPa with a rate of around  $2$  K  $\text{GPa}^{-1}$  (Fig. 1b). The highest pressure studied is 271 GPa, at which the material has  $T_c \approx 280$  K. A Pt lead inside the cell failed as the pressure was increased from 267 GPa, forcing the use of an adjacent Pt lead as a combined current–voltage probe (quasi-four-point measurement). We estimate the contribution from this shorted section of the Pt lead to be only  $\sim 0.1 \Omega$  (Extended Data Fig. 4). Additionally, no change in the shape of the superconducting transition was observed when the current was reduced to 0.1 mA, hence indicating bulk—rather than filamentary—superconductivity. These results were confirmed by a large number of experiments with over three dozen samples (see Supplementary Information and Extended Data Fig. 7). We note that the resistance of the sample decreases with increasing pressure, showing that it becomes more metallic at higher pressures.

## a.c. magnetic susceptibility

A superior test for superconductivity is the search for a strong diamagnetic transition in the a.c. magnetic susceptibility. In Fig. 2a, the real part of the temperature-dependent a.c. magnetic susceptibility  $\chi'(T)$  of the sample is shown for one of the experimental runs. The onset of superconductivity is signalled by a large ( $10$ – $15$  nV), sharp drop in susceptibility indicating a diamagnetic transition, which shifts to higher temperatures with increasing pressure. The highest transition temperature measured in this way is 198 K (transition midpoint), reached at the highest pressure measured (189 GPa). The quality of the data is high given the small sample size ( $\sim 80 \mu\text{m}$  in diameter and  $5$ – $10 \mu\text{m}$  in



**Fig. 2 | Magnetic susceptibility and superconducting transition under an external magnetic field.** **a**, Real part of the a.c. susceptibility in nanovolts versus temperature for the C–S–H system at select pressures from run 2, showing substantial diamagnetic shielding of the superconducting transition for pressures of 160–190 GPa. The superconducting transition shifts rapidly under pressure to higher temperatures.  $T_c$  is determined from the temperature

thickness). However, at extremely high pressures (>200 GPa), measuring the magnetic susceptibility becomes increasingly difficult, and impossible with sample diameters smaller than about 70  $\mu\text{m}$ . Our typical samples are about 25–35  $\mu\text{m}$  in diameter above 200 GPa. The substantial challenges to measuring properties such as the magnetic susceptibility suggest a need for novel experimental capabilities, such as spectroscopic techniques or magnetic sensing using nitrogen vacancy centres<sup>31–33</sup>.

### Magnetic-field response

To further confirm the superconducting transition at higher pressure we exploit the inherent antagonism between an external magnetic field and superconductivity. Within Bardeen–Cooper–Schrieffer theory, an external magnetic field exerts a Lorentz force to the opposite momenta of the electrons in a Cooper pair (the diamagnetic effect) and induces a Zeeman effect polarizing the initially spin-paired states of the pair electrons (the Pauli paramagnetic effect). Both of these effects result in the breaking of a Cooper pair, thus reducing the  $T_c$  of the material and setting an upper critical field,  $H_c$ , that the superconducting state can survive. In this study, the superconducting transition was suppressed by 22 K at 267 GPa in a 9-T magnetic field, as shown in Fig. 2b, confirming a superconducting transition. The transition was first measured at 210 GPa, followed by a second measurement at 267 GPa. The temperature dependence of the upper critical field,  $H_c(T)$ , can be expressed using the Ginzburg–Landau (GL) or the conventional Werthamer–Helfand–Hohenberg (WHH) model. Evaluating these relations at the limit of  $T = 0$  K at 267 GPa yields  $H_{c2}(0) = 61.88$  T with a coherence length of 2.31 nm for the GL model. From the WHH model, in the ‘dirty’ limit  $H_{c2}(0)$  can be extrapolated from the slope of the  $H$ – $T$  curve as  $H_{c2}(0) = 0.693 \left. \frac{dH_{c2}}{dT} \right|_{T=T_c} T_c$ , and this yields  $H_{c2}(0) = 85.34$  T (Fig. 2b, inset) with a coherence length of 1.96 nm. At 210 GPa,  $H_{c2}(0)$  and the coherence length at  $T = 0$  are 47.74 T and 2.63 nm and 66.18 T and 2.23 nm for the GL and WHH models, respectively (see Extended Data Fig. 3). The superconducting transition width,  $\Delta T_c$ , at 267 GPa remains essentially constant under several external magnetic fields, which is emblematic of a homogeneous sample;  $\Delta T_c = T_{90\%} - T_{10\%}$ , where  $T_{90\%}$  and  $T_{10\%}$  are the temperatures corresponding to 90% and 10% of the

resistance at 290 K. The background signal, determined from a non-superconducting C–S–H sample at 108 GPa, has been subtracted from the data. **b**, Low-temperature electrical resistance under magnetic fields of  $H = 0$  T, 1 T, 3 T, 6 T and 9 T (increasing from right to left) at 267 GPa. Inset, upper critical field versus temperature at 210 GPa and 267 GPa, fitted with the GL and WHH models. At 210 GPa, the maximum field studied was 7 T.

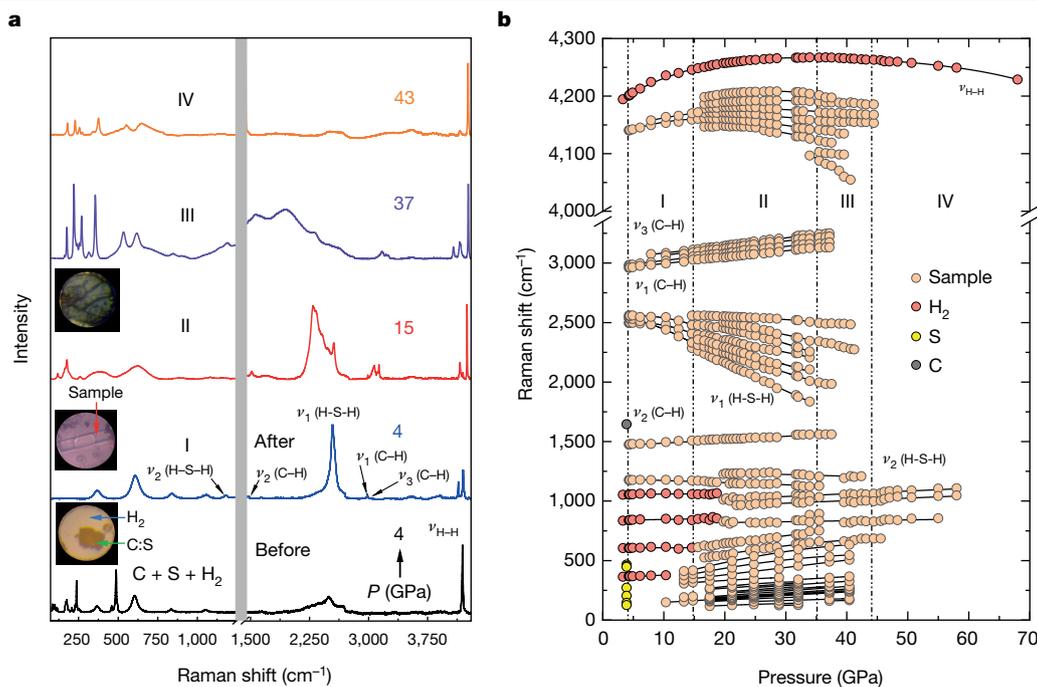
resistance at 290 K. The resistance  $R$  shows supralinear behaviour with respect to the temperature above the superconducting transition and follows  $R(T) = R_0 + AT^2$  with pre-factor  $A = 2.53 \times 10^{-4} \text{ m}\Omega \text{ mK}^{-2}$ , where  $R_0$  is the residual resistance; this behaviour can be described by inelastic electron–electron scattering within the Fermi liquid model (see Fig. 2b). At higher temperatures, one would typically anticipate an  $R(T) \propto T$  dependence according to the Bloch–Grüneisen law for a free-electron metal at temperatures well above the Debye temperature. The unusual behaviour indicates that the  $T^2$  term in  $R(T)$  is probably due to coupling to high-energy phonon modes, as is observed in  $\text{H}_3\text{S}$  (ref. 34).

### Photochemical synthesis

The starting compound is synthesized by combining elemental carbon and sulfur with a molar ratio of 1:1. The mixture is ball-milled to a particle size of less than 5  $\mu\text{m}$  and then loaded into a diamond anvil cell (DAC), after which molecular hydrogen is gas-loaded at 3 kbar to serve as both a reactant and a pressure-transmitting medium (PTM). Raman scattering confirmed the presence of the starting materials in the DAC. The confirmed DAC samples were compressed to 4.0 GPa and exposed to 532-nm laser light for several hours at a power of 10–25 mW. Irradiating the elemental sulfur phase ( $\alpha\text{-S}_8$ ) with light of this wavelength at these pressures is known to drive the photocission of S–S bonds, producing S free radicals, which either self-react to form different chain structures or, in this case, react with  $\text{H}_2$  to form  $\text{H}_2\text{S}$  (ref. 35). Slight adjustments were made in the pressure and laser position until the rapid formation of a uniform and transparent crystal that did not display Raman features from either elemental sulfur or  $sp^2$  carbon (see Supplementary Video). The molecular  $\text{H}_2\text{Q}_i(\text{J})$  vibron of the excess PTM was observed throughout, unperturbed and present up to the highest pressures. It is important to note that the crystal is not stable under 10 GPa, and exposure to low-intensity laser light or leaving it overnight at room temperature often caused the sample to disappear; however, we were able to collect Raman data at a few pressure points under 10 GPa.

### Raman spectroscopy before metallization

The Raman spectra of the transparent photoproduct formed at 4.0 GPa (Fig. 3) can be attributed to a H–S–H bending mode ( $\nu_2$ ), a S–H stretching



**Fig. 3 | Pressure-induced Raman changes of the photochemical product of C + S + H<sub>2</sub> mixtures. a**, Pressure-induced Raman changes of the photochemical product of C + S + H<sub>2</sub> mixture, showing the sulfur, carbon and fluid hydrogen (black) before the photochemical process at 4.0 GPa and spectral evidence of a photochemically transformed H<sub>2</sub>S + CH<sub>4</sub> + H<sub>2</sub> crystal at 4.0 GPa (blue). Note that no sulfur or carbon Raman modes are present in the sample after the photochemical process, indicating the synthesis of a new molecular compound. The insets show microphotographs of a synthesized transparent crystal at 4.0 GPa (I). At 15 GPa (II; red), the solid undergoes a phase transition marked by the appearance of new lattice modes, splitting of the H<sub>2</sub>S modes and

mode (ν<sub>1</sub>), a H-C-H bending mode (ν<sub>2</sub>), C-H stretching modes (ν<sub>1</sub>+ν<sub>3</sub>) and the splitting of the H<sub>2</sub> vibron, all of which together signify a H<sub>2</sub>S + CH<sub>4</sub> + H<sub>2</sub> van der Waals solid as described above. Compression above 10 GPa leads to the emergence of low-frequency lattice modes in the Raman spectra, which are indicative of increased periodicity in the sample. The I-II phase boundary at 15 GPa shows a progression similar to that of the guest-host structure reported by Strobel et al.<sup>7</sup>, including the emergence of the low-frequency lattice modes, the splitting of the H-S-H bending and S-H stretching modes, as well as a further splitting of the Q<sub>1</sub>(J) vibron of the guest H<sub>2</sub>. Thus, the phase transition can be described as a disorder-order transition akin to what is seen in (H<sub>2</sub>S)<sub>2</sub>H<sub>2</sub>, where the H<sub>2</sub>S molecular units align. In addition to the changes in the H<sub>2</sub>S-related modes, the splitting between the CH<sub>4</sub> modes increases, corroborating an ordering of an alloyed host framework. Regime III of the Raman spectra begins at 37 GPa, and it reveals the emergence of new lattice modes, further splitting of the H<sub>2</sub> vibron and the eventual disappearance of most Raman features. From these data, it can be concluded that another phase change occurs at 37 GPa. No Raman feature indicating the presence of sulfur is discernible, confirming that the carbonaceous sulfur hydride system does not decompose chemically (Extended Data Fig. 1). The Raman signal is lost above 60 GPa, and metallization is confirmed with transport measurements.

## Discussion

Although the ternary phase diagram is unknown, both H<sub>2</sub>S and CH<sub>4</sub> are known to form stable guest-host structures with H<sub>2</sub> under modestly high pressures<sup>7,36</sup>. The binary phase diagram for CH<sub>4</sub> + H<sub>2</sub> shows that a mixed fluid phase should form at our experimental photochemical

further splitting of the molecular H<sub>2</sub> vibron (ν<sub>1</sub>-H-H). Several new lattice modes appear above 37 GPa (III; purple), suggesting another phase transition. All the modes disappear above 60 GPa owing to the photoproduct undergoing an insulator-to-metal transition. **b**, Pressure versus Raman frequency up to 60 GPa, showing the spectral changes associated with the photochemistry of the C + S + H<sub>2</sub> mixture at 4 GPa, the phase transition of the photoproduct at 15 GPa and additional transitions at 37 GPa. The symbols correspond to the experimental data points, and the lines represent polynomial fits. The deconvoluted Raman peaks are shown in Extended Data Fig. 2.

synthesis conditions<sup>36</sup>; however, the observed Raman modes associated with C-H stretches do not evolve as for the free molecule, and the splitting of the ν<sub>1</sub> and ν<sub>3</sub> modes at 4.0 GPa clearly indicates that CH<sub>4</sub> is in the solid phase. We do not observe the shift to higher frequency of the molecular H<sub>2</sub> Q<sub>1</sub>(J) vibron, which is associated with the formation of (CH<sub>4</sub>)<sub>2</sub>H<sub>2</sub> or (CH<sub>4</sub>)<sub>2</sub>H<sub>2</sub> (ref. 36); this implies that the CH<sub>4</sub> is mixed with the H<sub>2</sub>S + H<sub>2</sub> guest-host structure that is known to form under these conditions (Extended Data Fig. 5). Therefore, the most likely compound to form at 4.0 GPa is a mixed alloy of H<sub>2</sub>S and CH<sub>4</sub> in the host framework, with stoichiometry (H<sub>2</sub>S)<sub>2-x</sub>(CH<sub>4</sub>)<sub>x</sub>H<sub>2</sub>. Ideally the stoichiometry is (H<sub>2</sub>S)(CH<sub>4</sub>)H<sub>2</sub> according to the composition of the starting materials. H<sub>2</sub>S and CH<sub>4</sub> have similar molecular kinetic diameters, they both form plastic face-centred cubic (fcc) phases with nearly identical lattice constants at 300 K and 4 GPa, and they both form Al<sub>2</sub>Cu-type guest-host frameworks of similar dimensionality with H<sub>2</sub> at 300 K and 4 GPa. These properties support the alloyed mixed van der Waals solid hypothesis. Of the 70 possible combinations that can be made from replacing four of the H<sub>2</sub>S molecules in the 4 GPa *I4/mcm* (H<sub>2</sub>S)<sub>2</sub>H<sub>2</sub> structure with CH<sub>4</sub> molecules (many of which will be identical by symmetry), we evaluated five with alternating molecules layered along [100], [010], [001] or [-110]. Four of those were stable to dissociation with respect to atomization and their molecular references, and the best one is shown in Extended Data Fig. 6. Structurally, lattice parameters *a* and *b* varied between 6.97 Å and 7.20 Å, *c* varied between 5.83 Å and 5.97 Å, and the lattice angles remained within 90.0° ± 0.8°.

Accurate structure determination of hydrogen-rich materials under very high pressure is extremely challenging, and we believe that the structural and stoichiometric determination of superhydride systems have been clouded by a reliance on X-ray diffraction (XRD) techniques.

Difficulties in probing lighter elements arise in XRD because the scattering power of an atom scales as  $Z^2$  ( $Z$ , atomic number). This causes heavier nuclei to dominate the signal, especially in materials such as rare-earth metal (Y, La) superhydrides. Consequently, reliable information regarding the location of the protons is lacking, with further complications in materials such as C–S–H, in which the overall scattering power of the sample is weak and the sample is single-crystal-like. To overcome such limitations, we are developing an alternative characterization suite of X-ray spectroscopy tools that provide information on the local electronic structure and coordination environment of a targeted element. Techniques such as X-ray absorption spectroscopy<sup>37</sup>, which probes the scattering from nearby atoms of X-ray-induced photoelectrons, and X-ray emission spectroscopy<sup>38</sup>, which provides an element-specific partial occupied density of states, allow a more complete structural picture of these hydrogen-rich materials than that afforded by XRD alone. These types of measurement in the megabar regime are currently extremely challenging, but we believe that they will enable the direct probing of elements such as carbon and sulfur. **‘Compositional tuning’ of these C–S–H ternary systems through controlling molecular exchange at lower pressures may be the key to achieving very-high- $T_c$  superconductors that are stable (or metastable) at ambient pressure. Therefore, a robust room-temperature superconducting material that will transform the energy economy, quantum information processing and sensing may be achievable.**

## Online content

Any methods, additional references, Nature Research reporting summaries, source data, extended data, supplementary information, acknowledgements, peer review information; details of author contributions and competing interests; and statements of data and code availability are available at <https://doi.org/10.1038/s41586-020-2801-z>.

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## Methods

### Sample preparation

A large number of high-pressure Raman spectroscopy, electric resistance, magnetic susceptibility and field-dependence resistance experiments were carried out, showing consistent reproducibility. The samples were loaded onto a membrane-driven DAC (m-DAC), using 1/3-carat, type Ia diamond anvils with a 0.2-, 0.15-, 0.1- and 0.05-mm (for higher pressures) culet with bevels of diameter of up to 0.3 mm at 8°. A 0.25-mm-thick rhenium gasket was pre-indented to 8–20  $\mu\text{m}$  (depending on the pressure) and a 120- $\mu\text{m}$ -diameter (or 70- $\mu\text{m}$ ; 30- $\mu\text{m}$  for high pressures) hole was electro-spark-drilled at the centre of the gasket. The starting compound was synthesized by combining elemental carbon and sulfur with a molar ratio of 1:1. The mixture was ball-milled to a particle size of less than 5  $\mu\text{m}$ . We loaded a clump of a C + S powder mixture into a DAC, and then molecular hydrogen was loaded to serve as both the reactant and the PTM. Raman scattering confirmed the presence of the starting materials in the DAC. The confirmed DAC samples were compressed to 4.0 GPa and exposed to 532 nm laser light for several hours at a power of 10–25 mW to synthesize the C–S–H crystals, which could grow as big as 80  $\mu\text{m}$  in diameter.

### Hydrogen loading and Raman spectroscopy

Highly compressed gases for loading into the m-DAC are necessary to obtain sufficient density. The density of gases at room temperature and atmospheric pressure is too low to obtain a sufficient quantity of sample after they condense into fluid or solid phases. A higher density provides enough hydrogen after collapsing the sample chamber as the pressure increases and offers a large enough sample to study. We used a high-pressure gas loader to compress gasses to high densities. The m-DACs were first loosely closed and mounted into a gearbox. Under a microscope, the DACs were opened 90° using the central gear of the gearbox. The DAC and gearbox were then placed into a high-pressure gas loader (Top Industries). The system was first flushed with hydrogen to purge the circuit of impurities. The sample chamber was pressurized to -2,500–3,000 bar. The gas loader was then drained, and the DAC was retrieved from the gearbox. In some experiments, high-purity hydrogen gas was cryogenically loaded into a DAC in a cryostat mounted on an optical table with  $\text{CaF}_2$  infrared-transmitting windows. To contain the liquid  $\text{H}_2$ , a mini chamber with a Balseal compressible O-ring was used<sup>39,39,40</sup>. A capillary tube was attached to the mini chamber for the gas flow. The optical table also incorporated standard instrumentation to measure Raman scattering and fluorescence.

Raman spectra were collected using a custom micro Raman setup using a 532-nm Millennia eV laser, a Princeton Instruments HRS 500 spectrometer and a Pylon camera. The laser was focused using a 20 $\times$  zoom objective (G Plan Apo 20X Objective). The reflected light travelled through a spatial filter and Bragg notch filters, allowing acquisition of low-wavenumber Raman peaks. Pressure determination was conducted using ruby fluorescence up to -110 GPa.  $\text{H}_2$  vibron ( $\nu_{\text{H-H}}$ ) frequency data were also used to determine the pressure in the range 58–196 GPa via the linear relationship  $P$  (GPa) =  $[4,392.559 - \nu_{\text{H-H}} (\text{cm}^{-1})]/2.277$ .

### Transport and magnetic susceptibility measurements

In all setups, the resistance in the two-probe configurations was measured using a Keithley DMM6500 multimeter, while the four-probe resistance was measured using a Keithley 2450 SMU and SIM921 a.c. resistance bridge<sup>13</sup>. In some experimental runs below -180 GPa, the superconducting transition was preceded by a sudden increase in resistance over the range of 230–200 K. It is not known whether this increase is associated with a charge- or spin-ordering transition. Whether or not the hump in  $R(T)$  around 200 K is indicative of a magnetic transition can only be answered through future temperature-dependent Mössbauer effect studies<sup>41</sup>. We used a custom-built BeCu DAC for magnetic-field-dependent electrical resistance studies using a physical

property measurement system. For the a.c. susceptibility experiments, the C–S–H sample was synthesized at 4.0 GPa via a photochemical process. We managed to grow crystals that covered almost the entire sample chamber. The DACs used type Ia diamonds with a 150- $\mu\text{m}$  culet with bevels up to 300  $\mu\text{m}$  at 8° diameter, and a MP35N gasket (neither superconducting nor magnetic) pre-indented to -12  $\mu\text{m}$  thick, into which a 120- $\mu\text{m}$  sample hole was drilled using an electric discharge machine. The coil had 180 turns with an inner diameter of -3.46 mm and a height of -1.95 mm. The details of the a.c. susceptibility measurements are discussed in ref. <sup>42</sup>. Superconductivity above 160 GPa appears at a somewhat lower  $T_c$  in the resistivity measurements than in the magnetic susceptibility ones, probably because of uncertainties in the estimated pressure at the sample due to pressure gradients.

### Structural optimization and enthalpy evaluations

We performed plane-wave density functional theory (PW-DFT)<sup>43,44</sup> calculations with the Vienna ab initio simulation package (VASP) version 5.4.4. The projector-augmented wave<sup>45</sup> pseudo-potentials formulated for Perdew–Burke–Ernzerhof (PBE) GW simulations<sup>46</sup> were used with the generalized gradient approximation PBE functional<sup>47</sup> using Grimme's D3<sup>48</sup> semi-empirical dispersion correction with Becke–Johnson damping. An automatically generated  $\Gamma$ -centred  $k$ -point mesh with spacing  $0.05 \times 2\pi \text{ \AA}^{-1}$  was used in all PW-DFT calculations with plane waves cut off at 500 eV. The Kohn–Sham<sup>44</sup> equations were solved using the RMM-DIIS algorithm. To minimize the effect of the Pulay stress, the constant-volume optimizations were performed in three parts: two sequential optimizations and a final single-point energy evaluation<sup>48–53</sup>. The energy convergence criterion for a self-consistent field cycle was  $10^{-8}$  eV, and a force convergence criterion of  $10^{-2}$  eV  $\text{\AA}^{-1}$  or 200 (at minimum) iterations was used to stop the geometry optimizations. The computed values for enthalpies were not vibrationally corrected because they were only intended as a rough screening of candidate compounds that initially form at 4.0 GPa.

### Simulations of initial photochemical products at 4.0 GPa

At around 4 GPa, both  $\text{CH}_4$  and  $\text{H}_2\text{S}$  are disordered solids in phase I, wherein the heavy atoms occupy an fcc lattice ( $a_{\text{CH}_4} = 5.1963 \text{ \AA}$ ,  $a_{\text{H}_2\text{S}} = 5.19503 \text{ \AA}$ )<sup>54,55</sup> and the H atoms are fully rotationally disordered about the heavy atom. A  $2 \times 2 \times 2$  fcc supercell of the heavy atoms was created, and the H atoms were placed by randomly rotating a pre-defined molecule on a unit sphere with bond lengths and angles  $r_{\text{S-H}} = 1.34 \text{ \AA}$ ,  $\phi_{\text{H-S-H}} = 92.0^\circ$ ,  $r_{\text{C-H}} = 1.09 \text{ \AA}$  and  $\phi_{\text{H-C-H}} = 109.5^\circ$ , respectively. The H positions were then allowed to optimize in VASP, keeping the volume and heavy atoms fixed. Because the structures did not find an actual minimum, owing to the flat potential surface for the rotation of the molecules, the enthalpy for each simulation volume was taken as the average of the final 20 optimization cycles.

Hydrogen at 4 GPa and 300 K is a molecular fluid with a density determined from sound velocity measurements<sup>56</sup>. The simulation volume was thus a cubic box of length 11.994581  $\text{\AA}$  with 112  $\text{H}_2$  molecules. The configuration space for the fluid was sampled with a classical NVT ( $T = 300 \text{ K}$ ) Monte Carlo process wherein the  $\text{H}_2$  molecules were at a fixed bond length ( $r_{\text{H-H}} = 0.741 \text{ \AA}$ ) and were allowed to rotate, translate and re-insert with equal probability. The classical  $\text{H}_2$  interactions were modelled with the Darkrim–Levesque<sup>57</sup> force field, including electrostatics and Feynman–Hibbs-corrected Lennard–Jones potentials. Ten snapshots were taken from the last 10,000 Monte Carlo cycles and passed into VASP to calculate the enthalpy.

The structures evaluated here for the 4 GPa, 300 K molecular van der Waals guest–host compounds of  $(\text{H}_2\text{S})_2\text{H}_2$ ,  $(\text{CH}_4)_2\text{H}_2$  (ref. <sup>58</sup>) and  $(\text{H}_2\text{S})(\text{CH}_4)\text{H}_2$  are the previously proposed  $I4/mcm$  and  $P\bar{1}$  ones of Strobel et al.<sup>7</sup> and Duan et al.<sup>6</sup> for  $(\text{H}_2\text{S})_2\text{H}_2$ . The other proposed  $(\text{H}_2\text{S})_2\text{H}_2$  phases ( $I222$ ,  $Cccm$ ,  $R\bar{3}m$  and  $Im\bar{3}m$ ) were not favourable for  $(\text{H}_2\text{S})_2\text{H}_2$  at these conditions and were thus ignored for  $\text{CH}_4$ -containing variants. The  $I222$  and  $P\bar{1}$  structures can be viewed as lowered-symmetry versions of

the *I4/mcm* structure, which is an Al<sub>2</sub>Cu archetypal in which where rotationally disordered H<sub>2</sub>S (or CH<sub>4</sub>) molecules occupy the Al sites and rotationally disordered H<sub>2</sub> molecules occupy the Cu sites. For the CH<sub>4</sub> variants, some or all of the H<sub>2</sub>S molecular units of the (H<sub>2</sub>S)<sub>2</sub>H<sub>2</sub> structure were replaced with the pre-determined CH<sub>4</sub> molecules and randomly rotated as described above. The H positions of the molecular H<sub>2</sub>S units in the reported *I4/mcm* structure of (H<sub>2</sub>S)<sub>2</sub>H<sub>2</sub> seem too highly symmetric. Those H atoms probably have rotational disorder about the heavy atom, as with phase I of H<sub>2</sub>S, as can be inferred from a recent Raman study that showed a I-to-II-phase change of (H<sub>2</sub>S)<sub>2</sub>H<sub>2</sub>, which we believe to be an ordering of the rotationally disordered H<sub>2</sub>S units of the *I4/mcm* structure to an ordered host (H<sub>2</sub>S)<sub>2</sub> sub-lattice, based on the analogy to the Raman data regarding the phase progression in pure H<sub>2</sub>S(s) (ref.<sup>59</sup>). In addition, it was recently shown that H<sub>2</sub>S does exhibit (at least weak) hydrogen bonding with an H...S distance of 2.779(9) Å in the dimer<sup>60</sup>, and the original proposed *I4/mcm* structure would violate the well known Bernal–Fowler ‘ice rules’ for a hydrogen-bound solid<sup>61</sup>. To compensate for this disorder, the H<sub>2</sub>S units in the reported structure were replaced by randomly rotated pre-determined H<sub>2</sub>S molecular units. This modification provides lower-enthalpy structures over the overly symmetrized published structure. Five variants of (H<sub>2</sub>S)(CH<sub>4</sub>)H<sub>2</sub> that replace four of the H<sub>2</sub>S units of *I4/mcm* (H<sub>2</sub>S)<sub>2</sub>H<sub>2</sub> were evaluated. They each appear stable against dissociation to the molecular species if the vibrational energies are not evaluated, although the pure phase of (H<sub>2</sub>S)<sub>2</sub>H<sub>2</sub> appears more stable; see Supplementary Table 3. Further work is under way to further understand the initial product by accurately quantifying its enthalpy and stoichiometry, as well as understanding its phase progression with increased pressure.

## Data availability

The data supporting the findings of this study are available within the article and its Supplementary Information files, and from the corresponding author upon reasonable request.

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**Author contributions** E.S., N.D.-G. and R.M. performed the Raman and electrical conductivity measurements and contributed to the writing of the paper; K.V. participated in the Raman measurements and analysis of the Raman data. H.V. analysed the low-temperature magnetic-field-dependent electrical conductivity measurements and contributed to the writing of the paper. M.D. provided technical support during the initial stage of the electrical conductivity measurements, performed magnetic susceptibility measurements and contributed to the writing of the paper. R.P.D. conceived the project and performed electrical conductivity and magnetic susceptibility experiments. K.V.L. and A.S. analysed the data and the chemistry protocol. K.V.L., A.S. and R.P.D. wrote the paper. All authors discussed the results and commented on the manuscript.

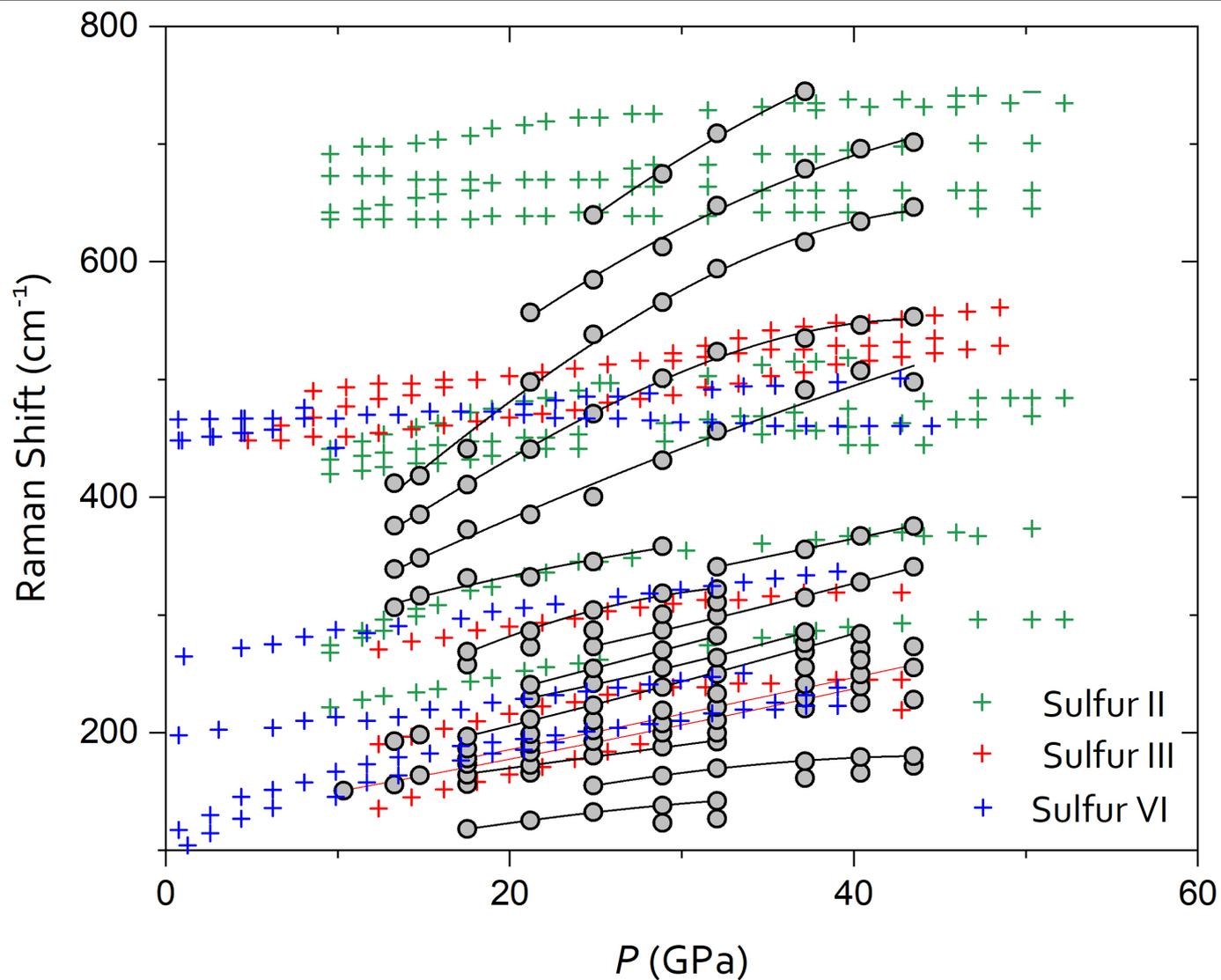
**Competing interests** The authors declare no competing interests.

## Additional information

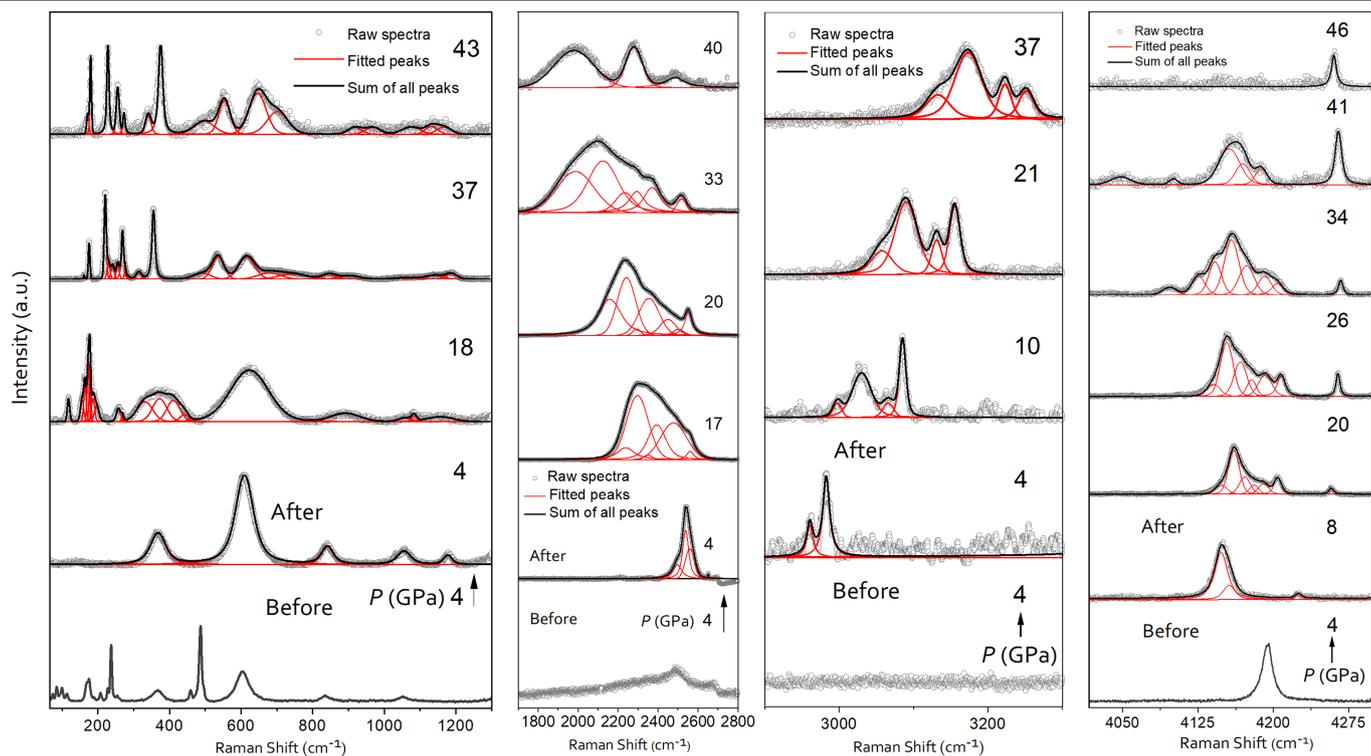
**Supplementary information** is available for this paper at <https://doi.org/10.1038/s41586-020-2801-z>.

**Correspondence and requests for materials** should be addressed to R.P.D.

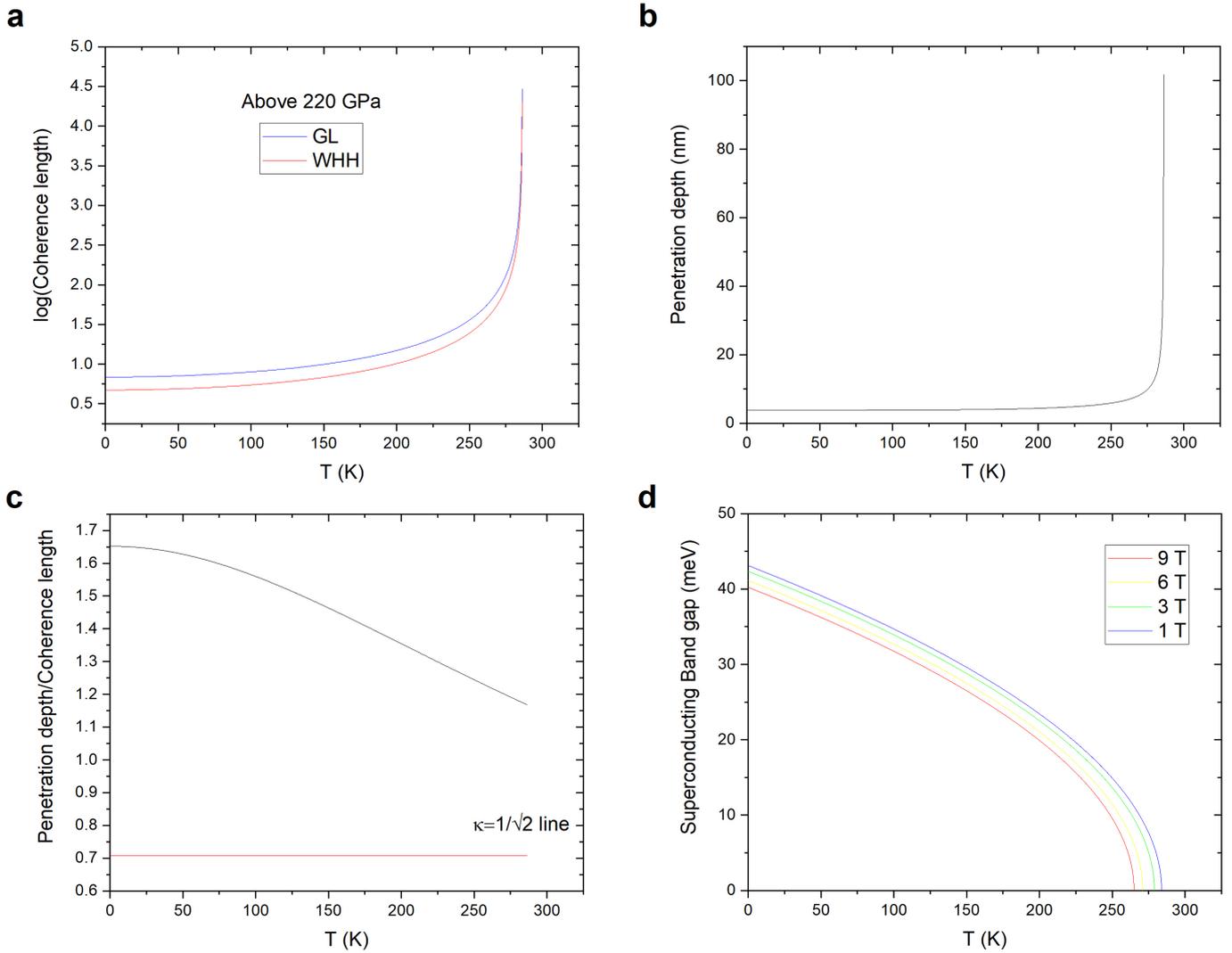
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**Extended Data Fig. 1 | Vibrational frequencies of the C-S-H sample.** Pressure versus Raman shift for low-frequency modes compared to pure sulfur modes. Grey solid markers are from the sample; the green, red and blue crosses represent different phases of pure sulfur.

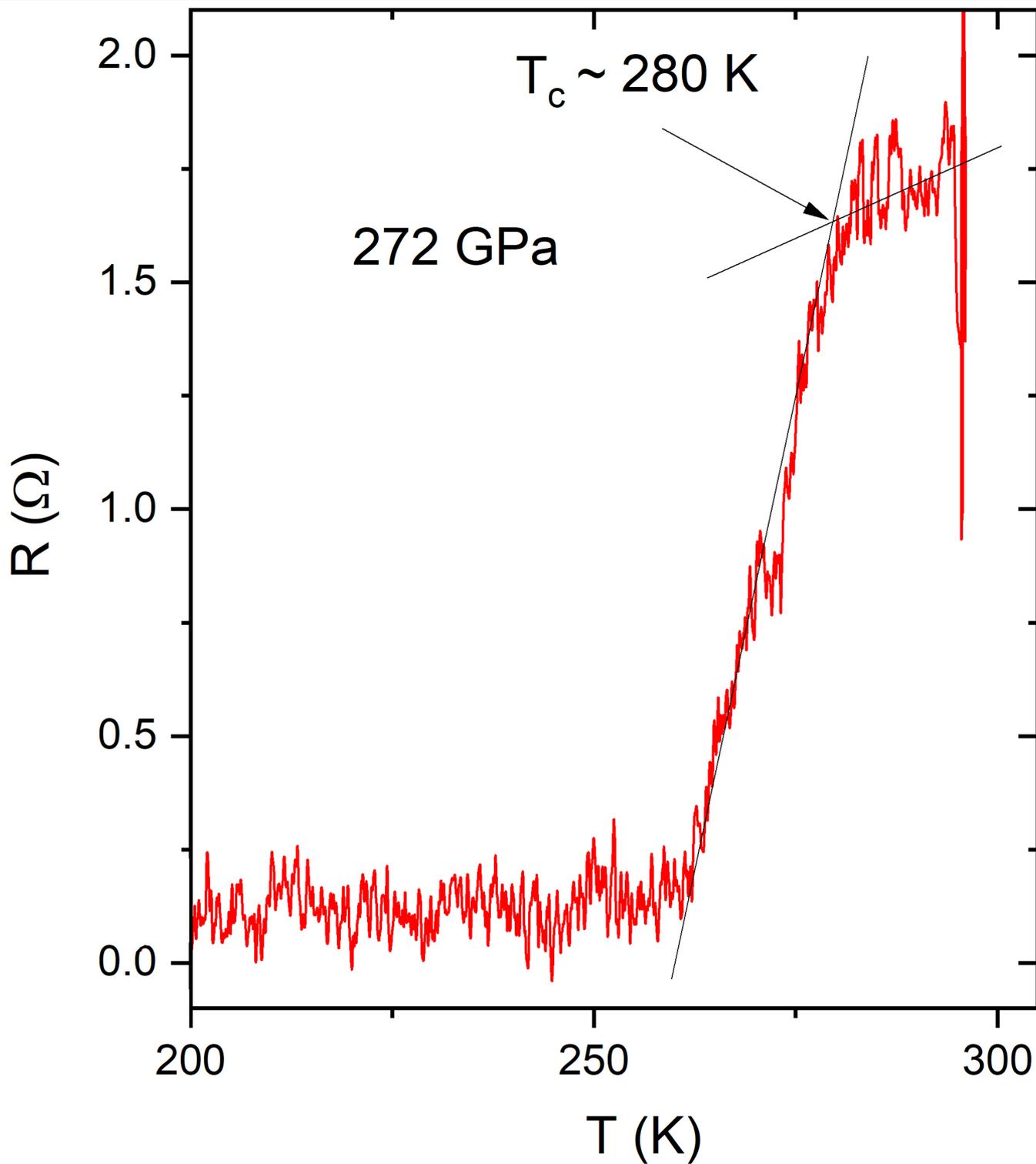


**Extended Data Fig. 2 | Raman spectra of the C-S-H sample.** Spectral deconvolution of Raman spectra of the C-S-H compound used to find the peak positions reported in Fig. 3b. a.u., arbitrary units.



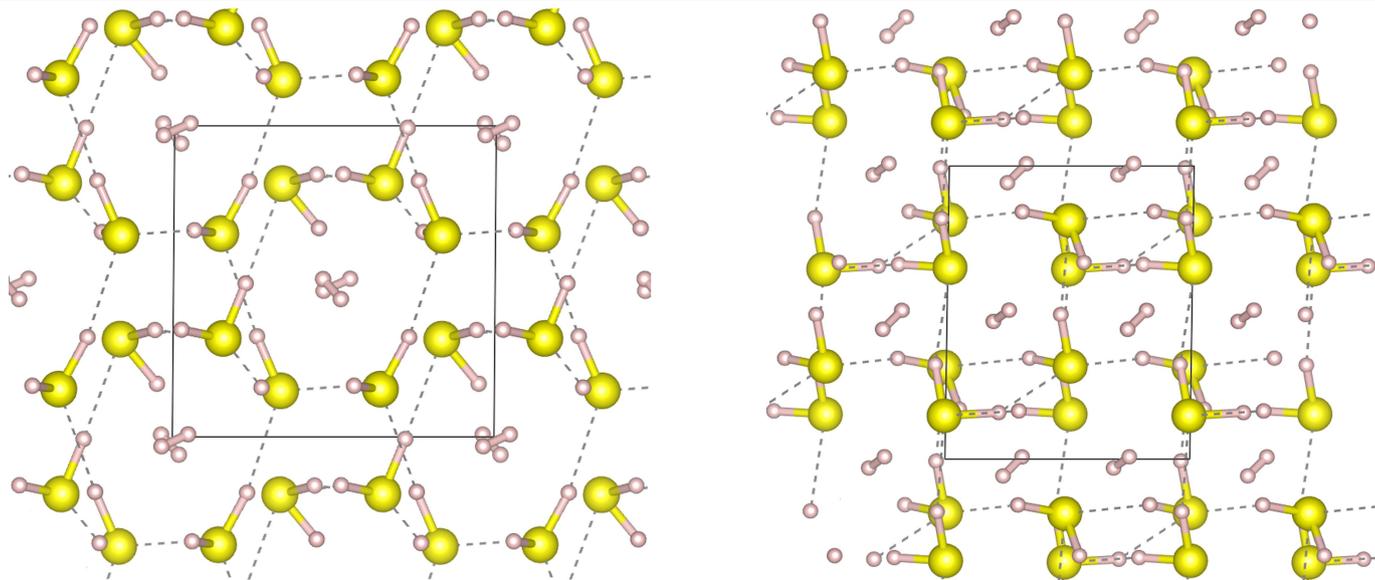
**Extended Data Fig. 3 | Superconducting properties of the C-S-H system.**  
**a**, Coherence length versus temperature. **b**, Temperature dependence of the penetration depth. The penetration depth can be determined to be  $\lambda(0) > 3.8$  nm as  $\lambda(0) = \phi_0 / [2\sqrt{2} \pi H_c(0) \xi(0)]$  with  $H_c(0) = 61.8$  T and  $\xi(0) = 2.3$  nm;  $\phi_0$  is the magnetic flux quantum. **c**, GL parameter,  $\kappa = \lambda(T) / \xi(T)$  at  $T = 0$ ;  $\kappa > 1/\sqrt{2}$

indicates type II superconductivity. In our experiment  $\kappa > 1.1$ , so the sample can be identified as a type II superconductor. **d**, Variation of the superconducting bandgap  $\Delta$  versus temperature,  $\Delta(T) / \Delta(0) = 1.76 k_B T_c \sqrt{1 - (T/T_c)}$  ( $k_B$ , Boltzmann constant).

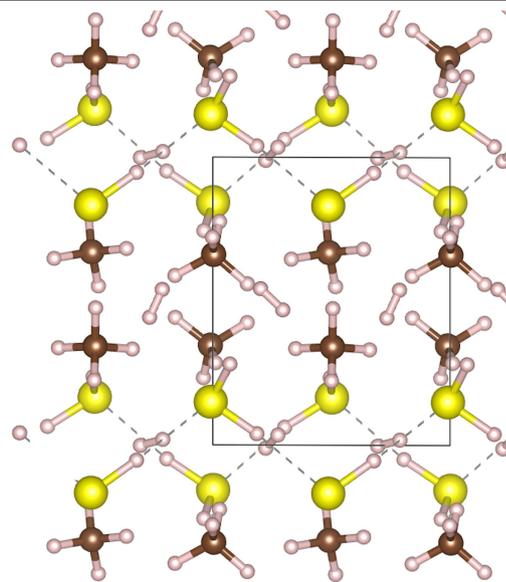
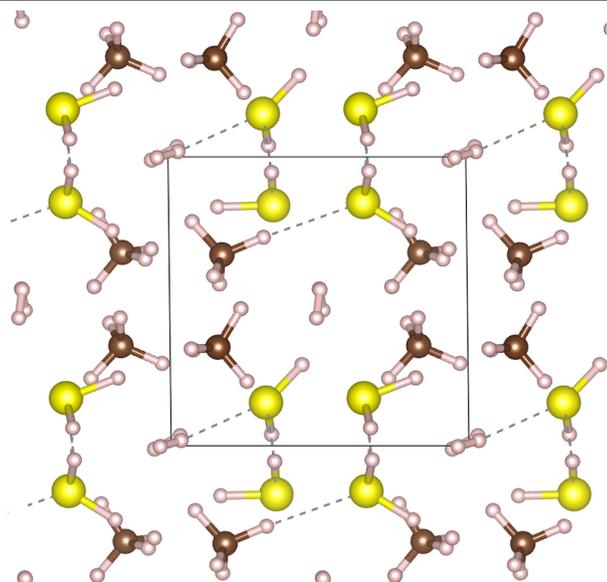


**Extended Data Fig. 4 | Superconducting transition at 272 GPa.** Temperature-dependent quasi-four-point electric resistance measurement of the C-S-H sample at 272 GPa, showing the superconducting transitions at  $\sim 280$  K.  $T_c$  was

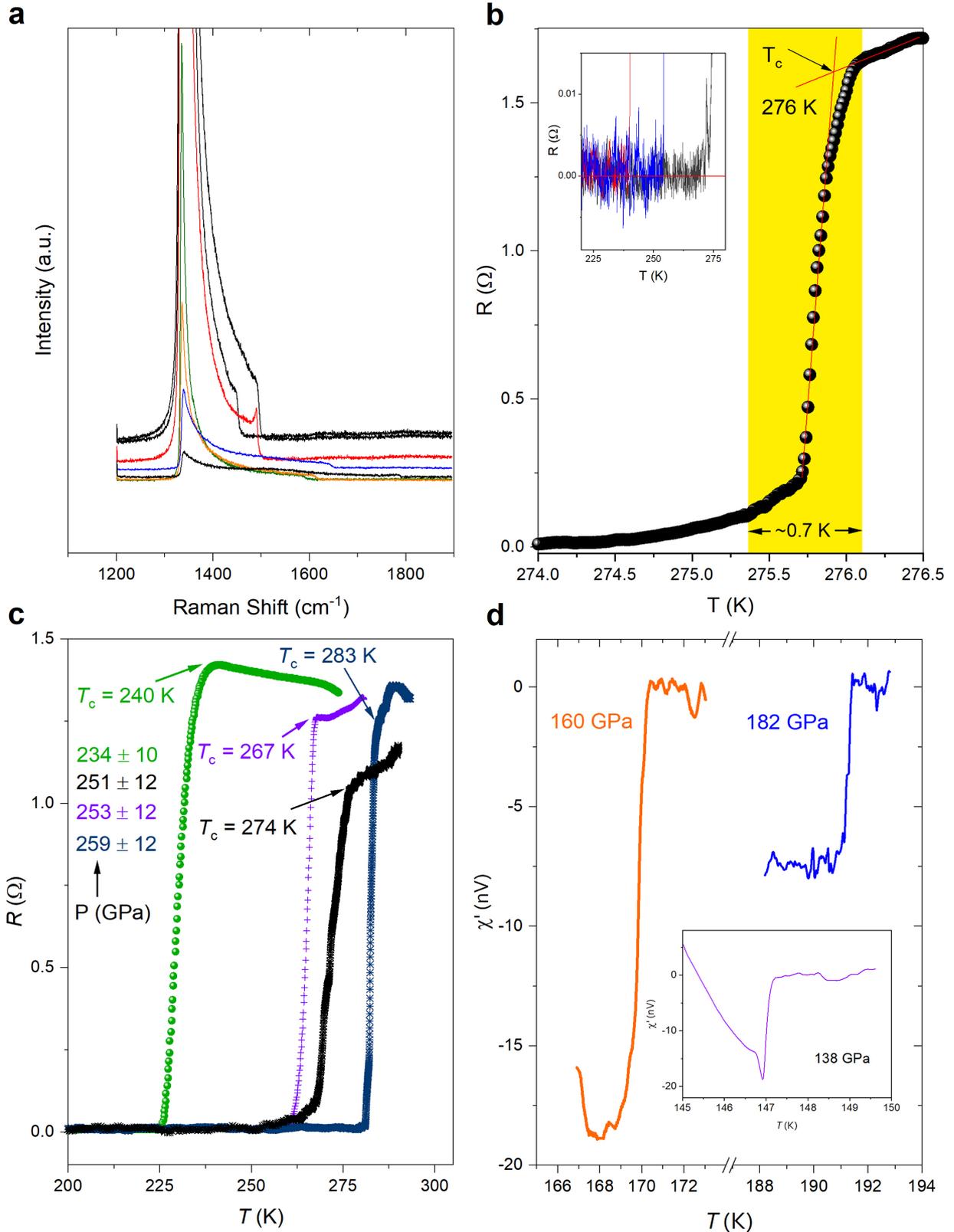
determined from the onset of superconductivity (see arrow). A superconducting step is observed at the transition midpoint.



**Extended Data Fig. 5 | ADF-optimized structure for  $(\text{H}_2\text{S})_2\text{H}_2$  at 4 GPa.** View along the (00-1) (left) and (-100) (right) planes of a calculated snapshot of  $(\text{H}_2\text{S})_2\text{H}_2$ . S-H distances between 1.4 Å and 2.8 Å are shown as dashed lines to indicate potential hydrogen bonds.



**Extended Data Fig. 6 | ADF-optimized structure for  $(\text{H}_2\text{S})(\text{CH}_4)\text{H}_2$  (variant 2) at 4 GPa.** View along the (00-1) (left) and (-100) (right) planes of a calculated snapshot of a variant (number 2) of  $(\text{H}_2\text{S})(\text{CH}_4)\text{H}_2$ . S-H distances between 1.4 Å and 2.8 Å are shown as dashed lines to indicate potential hydrogen bonds.



**Extended Data Fig. 7 | Additional data on superconducting transition and pressure measurements.** **a**, Representative spectra of the diamond Raman edge used for pressure determination. **b**, Plot of resistance versus temperature, showing that the width of the superconducting drop is less than

1 K. Inset, zoom-in of the resistance near  $0 \Omega$ . **c**, Temperature dependence of the resistance of C-S-H at different pressures (different samples). **d**, Temperature dependence of the measured a.c. susceptibility of C-S-H at 182 GPa. The inset shows raw data at 138 GPa.